

BRIEF COMMUNICATION

Studies on Compensated Cu-Cr-Al Spinel Oxide Semiconductors

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Electrical resistivity (ρ) and thermoelectric power (α) measurements were carried out in air on $\text{CuCr}_{2-x}\text{Al}_x\text{O}_4$ ($0.06 \leq x \leq 0.10$) spinel oxides in the temperature range 300–675 K. Carrier concentration (n), optical phonon frequency (γ_0), and mobility (μ) values have been calculated from resistivity and thermoelectric power data. The results show that $\text{CuCr}_{2-x}\text{Al}_x\text{O}_4$ spinel oxides are compensated semiconductors and the total compensation of the hole carriers by electrons occur when 0.09 chromium ions are replaced by aluminium ions. © 1995 Academic Press, Inc.

INTRODUCTION

The spinel oxide CuCr_2O_4 is a p -type semiconductor. Earlier studies have shown that on substituting Cr^{3+} by Al^{3+} the samples, $\text{CuCr}_{2-x}\text{Al}_x\text{O}_4$, become n -type (1). The present work seeks to determine the minimum amount of Al^{3+} , i.e., x , required to change p -type CuCr_2O_4 to an n -type semiconductor.

EXPERIMENTAL

The solid solutions $\text{CuCr}_{2-x}\text{Al}_x\text{O}_4$ for $x = 0.06, 0.08, 0.09$, and 0.10 were prepared from stoichiometric mixtures of the respective nitrates by the method described in an earlier paper (2). X-ray diffraction studies of all the samples were carried out using a Philips X-ray diffraction unit (Model PW1710) using $\text{CuK}\alpha$ radiation with Ni filter. Resistivity and thermoelectric power measurements of all the samples were carried out in air in the temperature range 300–675 K following the procedure described elsewhere (1, 3). All the samples were annealed before each measurement to eliminate a grain boundary effect.

RESULTS AND DISCUSSION

Table 1 gives the chemical composition of various aluminium-substituted CuCr_2O_4 samples, their code names,

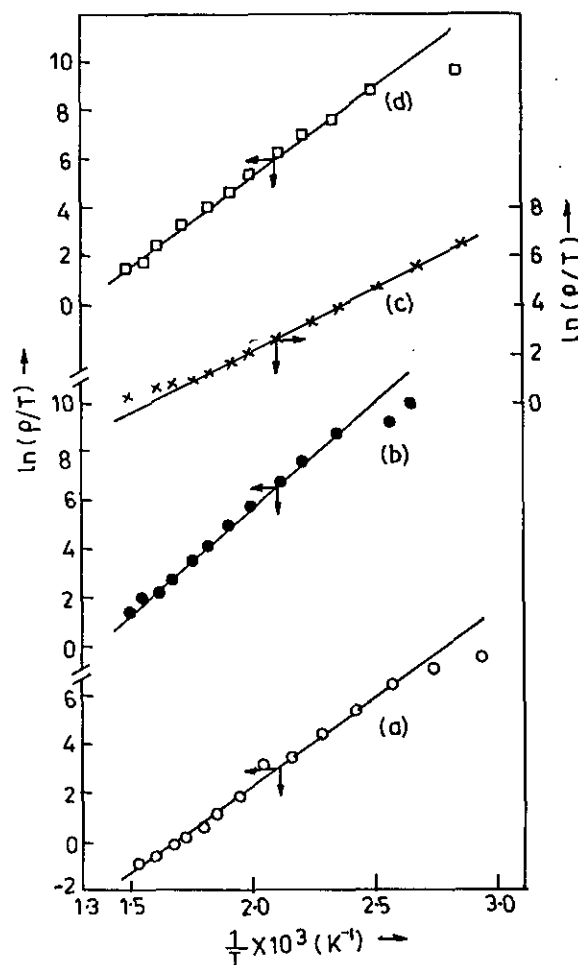


FIG. 1. Plots of $\ln(\rho/T)$ vs $1/T$ for (a) NA6, (b) NA8, (c) NA9, (d) NA10.

optical phonon frequency (γ_0), and mobility (μ) values at 550 K. Resistivity of all the samples followed the exponential law

$$\rho = \rho_0 \exp(E_a/kT),$$

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TABLE 1
Sample Composition, Code Name, Optical Phonon Frequency
(γ_0), and Mobility (μ) for Various $\text{CuCr}_{2-x}\text{Al}_x\text{O}_4$ Samples

| Sample composition | Code name | γ_0 (sec^{-1}) | $\mu_{550\text{K}}$ ($\text{cm}^2\text{V}^{-1}\text{sec}^{-1}$) |
|--|-----------|----------------------------------|---|
| $\text{CuCr}_{1.94}\text{Al}_{0.06}\text{O}_4$ | NA6 | 1.19×10^{12} | 2.5×10^{-7} |
| $\text{CuCr}_{1.92}\text{Al}_{0.08}\text{O}_4$ | NA8 | 7.99×10^{12} | 1.43×10^{-7} |
| $\text{CuCr}_{1.91}\text{Al}_{0.09}\text{O}_4$ | NA9 | 5.41×10^{10} | 1.96×10^{-7} |
| $\text{CuCr}_{1.90}\text{Al}_{0.10}\text{O}_4$ | NA10 | 2.42×10^{11} | 2.64×10^{-8} |

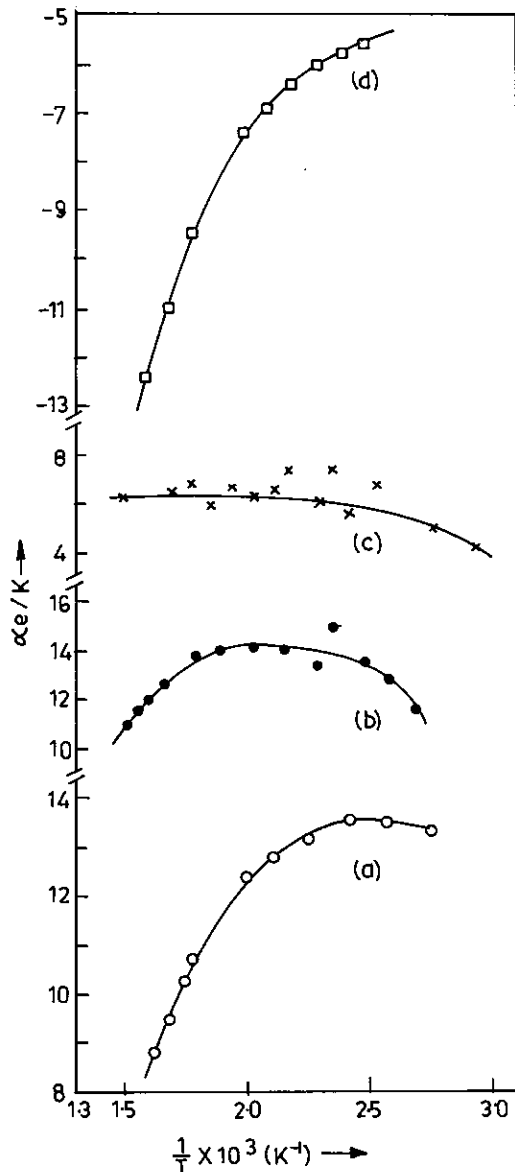


FIG. 2. Plots of α_e/k vs $1/T$ for (a) NA6, (b) NA8, (c) NA9, (d) NA10.

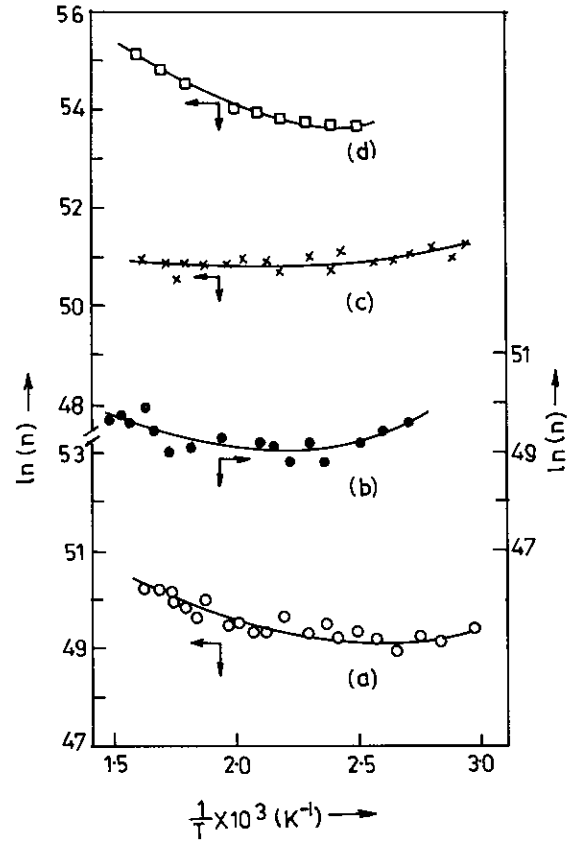


FIG. 3. Plots of $\ln(n)$ vs $1/T$ for (a) NA6, (b) NA8, (c) NA9, (d) NA10.

in the temperature range studied, where ρ is resistivity in ohm-cm, T the absolute temperature, k the Boltzmann constant, and E_a the thermal activation energy. The $\ln(\rho/T)$ vs $1/T$ plots for all the samples are shown in Fig. 1. From the $\ln(\rho/T)$ vs $1/T$ linear plots, the optical phonon frequency values (γ_0) and mobility values (μ) have been calculated following the method of Metselaar *et al.* (4).

The reduced thermoelectric power α' ($=\alpha_e/k$) vs $1/T$ plots for all the samples are shown in Fig. 2. The results show that NA6 is a p -type semiconductor in the temperature range studied whereas NA10 is n -type in the same temperature range. NA8 and NA9 appear n -type at room temperature but on heating change to p -type. Following Basak and Ghose (5), the values of carrier concentration n have been calculated from the values of α using the formula

$$n = Nv \exp(-\alpha/k),$$

where Nv is the density of states involved in the conduction process, k is the Boltzmann constant, and α is the Seebeck coefficient. The plots of $\ln(n)$ vs $1/T$ are shown in Fig. 3. The plots show that for all the samples except NA9, the carrier concentration remains almost constant up to 500 K

and then on heating increases. NA9, however, does not show any change in the carrier concentration in the temperature range studied.

The results show that the $\text{CuCr}_{2-x}\text{Al}_x\text{O}_4$ samples with $x \leq 0.06$ are *p*-type at room temperature and $x \geq 0.08$ are *n*-type. The linear $\ln(\rho)$ vs $1/T$, $\ln(\rho/T)$ vs $1/T$, and the calculated γ_0 values indicate that all samples are semiconductors where conduction occurs by hopping of charge carriers. Murthy and co-workers (3) have shown that CuCr_2O_4 is *p*-type due to the presence of Cr^{4+} ions in the spinel oxide. From ESCA measurements Padmanaban *et al.* (1) have shown that the Al^{3+} substituted samples contain some Cu^{1+} ions and when these exceed the Cr^{4+} ions present in CuCr_2O_4 , the samples become *n*-type and hence $\text{CuCr}_{2-x}\text{Al}_x\text{O}_4$ spinel oxides were considered to be compensated semiconductors. In the present studies the carrier concentrations, n , for all the samples were calculated. Figure 3 shows that in NA6 the charge carrier concentration initially remains constant and then increases above 500 K;

NA8 shows an initial fall and then n increases slightly above 550 K. NA9 shows almost no change in n after an initial fall and in NA10, n increases with increasing temperature. These results conclusively show that the $\text{CuCr}_{2-x}\text{Al}_x\text{O}_4$ spinel oxides are compensated semiconductors and the total compensation of the hole carriers by electrons occur when 0.09 chromium ions are replaced by aluminium ions.

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